

Research Article

Conductive Polymeric Composites Based on Multiwalled Carbon Nanotubes and Linseed Oil Functionalized and Cross-Linked with Diacetylenes from Propargyl Alcohol

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Diacetylene-functionalized epoxidized linseed oil (DAELO) matrix was synthesized in order to improve the dispersion of multiwalled carbon nanotubes (MWCNTs) without the necessity of some chemical or physical modifcation of them. Tat fact was evidenced by the low critical concentration of DAELO-based composites in comparison (1.0wt% MWCNTs) with the epoxidized linseed oil- (ELO-) based composites (5wt% MWCNTs). For this, both series of composites were prepared by the ultrasonic dispersion method using the same conditions of solvent, dilution, and sonication time. It was shown that, tailoring the polymer matrix with groups rich in nonpolar electric density, as diacetylene, and capable of interacting by van der Waals forces, it is possible to improve the dispersion of carbon nanotubes (CNTs) without necessity of some modifcation knowing that those treatments usually afect lowering their electrical properties.

1. Introduction

Carbon nanotubes (CNTs) have excellent electrical and mechanical properties [1, 2], which makes them ideal candidates for several applications, for example, as nanofller material in conductive polymeric compounds [3]; CNTs are quite efective compared to carbon black microparticles, due to the large aspect ratio. Incorporation of CNTs into polymeric compounds gives rise to new opportunities for electronic applications, for example, in conductive flms [4, 5], photovoltaic devices $[6]$, or sensors $[7-9]$. Their advantages are that they may be fexible, easy to mold, and cheap; nonetheless they have the drawbacks of the polymers from fossil sources. In this context, polymers from natural sources, for example, from vegetable oils, represent a good alternative for the cost, environment, and chemical transformation, making them tailored structures [10–12]. As an example, linseed oil is a triglyceride consisting of three fatty acids condensed onto a glycerol unit; its composition is mainly of linolenic acid (56.6%), linoleic acid (15.3%), and oleic acid (19.1%) moieties [13] (Figure 1), so it has on average 6.4 double bonds per triglyceride unit [14]; this feature makes it easy to modify by chemical conversion to other functional groups as epoxides. Afer the epoxidation, these groups could be used as cross-linker site or could be chemically modifed in order to have a specifc group in their structure [15,16]. For an efective and reproducible utilization of CNTs in these composites, it is very important to have a good and homogeneous dispersion of the conductive particles throughout the polymer matrix. However, this is likely the main drawback owing to van der Waals forces between them along with their structure; they form clusters or tight long bundles and these might form a dense and entangled network; therefore in order to improve their dispersion, surface modifcations have been made [17, 18]; one of them is the noncovalent attachment of molecules although their disadvantage is that the forces between the molecules and the CNTs might be weak; another one is the covalent attachment of functional groups to the walls of

Figure 1: Chemical structure and main composition of linseed oil (LO).

CNTs; however this decreases the conductivity because the groups might introduce defects on the graphitic structure [19, 20]. Functionalization with divalent adducts represents an option which alters less the electrical transport owing to unaltered $sp²$ conjugation [21]. Other options have been the dispersion by shear mixing, using surfactants [22], or by sonication in diferent solvents [23, 24]. In order to avoid any modifcation to the CNT and some loss of the electrical properties, we thought in synthesizing a matrix with functional groups rich in electrical density without dipole moment, as the diacetylenes, but that it may interact with the electrical density of the CNTs via van der Waals forces, assisting the dispersion.

The proposal was the functionalization of the epoxidized linseed oil with propargyl alcohol by a catalytic ring opening reaction in order to have terminal acetylenes groups (APELO), followed by an oxidative coupling reaction C– C to form diacetylene groups which also serve as crosslinkers of the oil. Finally, solutions of DAELO and MWCNTs at diferent loads were mixed using an ultrasonic bath, the percolation curve was built, and the critical concentration for diacetylene linseed oil was lower than ELO, demonstrating that rich electron groups have a positive efect on the dispersion of MWCNTs as compared with ELO.

2. Materials and Methods

Lipase acrylic resin from *Candida antarctica*, MWCNTs $D \times$ L 110–170 nm \times 5–9 μ m, density 1.7–2.1 g/cm³, linseed oil (LO), hydrogen peroxide, alumina, anhydrous zinc chlorine, propargyl alcohol (PA), copper chlorine, and tetramethyl ethylene diamine (TMEDA) were obtained from Sigma Aldrich Co.; sodium carbonate and magnesium sulphate anhydrous were obtained from J. T. Baker, Mexico. All the reactants were used as received. SEM images were obtained in a JEOL JSM-6510LV microscope at acceleration voltage of 30 kV, at 5000, 10000, and 20000x, with backscattered electrons detector. ¹H NMR spectra were recorded at room temperature using $CDCl₃$ as solvent on a Bruker Avance 300 MHz NMR. Chemical shifts are relative to $(CH_3)_4$ Si and are given in ppm. FTIR-ATR spectra were recorded on an Avatar FTIR-ATR spectrophotometer, into a range between 550 and 4000 cm⁻¹ and spectra were acquired at 4 cm^{-1} resolution and signals averaged over 32 scans. DSC and TGA analysis were recorded under nitrogen atmosphere (100 mL/min) using a SDT-Q600 TA Instruments modulus from 20 to 600[∘] C and heating rate of 20[∘] C/min. Electrical resistance was measured on a digital multimeter ASYC II 5390.

ELO was synthesized using the chemoenzymatic method described in previous report [25]; a toluene solution, 100 g of LO, 8 g of oleic acid, and 10 g of lipase were put into a precipitate glass using 120 mL of solvent; the reaction mixture was warmed at 42[∘] C and mixed by stirring at 355 rpm; then 160 mL of hydrogen peroxide at 30% drop by drop was added. Afer 24 hours, the reaction was fltered to recover the enzyme. The product was purified by successive extractions of 3×20 mL of Na₂CO₃ solution at 10% and 1×20 mL of water; finally ELO was dried using $MgSO₄$ anhydrous. The product was characterized by FTIR-ATR, ¹H NMR, DSC, and TGA.

For synthesis of functionalized ELO with propargyl alcohol (PAELO), a mixture of 0.5 g of ELO, ZnCl₂ at different percentages (5–20%), and 0.5 mL of propargyl alcohol was heated at diferent temperatures (60–90[∘] C) and times (3– 20 h); a change of color was observed from white to amber or red. PA and ZnCl₂ were extracted with water; the product was fltered and solved in acetone; fnally it was dried by a rotary evaporator and by a vacuum line. These conditions were carried out in order to found those that render the maximal functionalization without side reactions. Products were characterized by FTIR-ATR, ¹H NMR, DSC, and TGA.

In order to form the diacetylene groups, a coupling C–C of terminal acetylenes was carried out at room temperature by the Hay reaction [26, 27]. PAELO was dissolved in chloroform, while an aqueous solution of CuCl and TMEDA in stoichiometric ratio 1:1 was prepared; both solutions were placed together and mixed by magnetic stirring under an air flow. After 40 min the reaction was stopped and a solid was observed; the organic solvent was evaporated. The catalyst obtained in situ Cu(TMEDA) [28] was extracted with water many times as necessary until the characteristic blue color was not observed. The final product was a yellow solid which was dried and characterized by FTIR-ATR, DSC, TGA, and UV-Vis spectroscopy.

The diacetylene-polymer composites (DAELO-MWCNTs) were prepared in three steps: frst a polymer solution of 10 mg/mL in chloroform was made and MWCNTs at concentrations between 0.1 and 1.2 mg/mL were dispersed in chloroform by an ultrasonic bath for 30 minutes at 35 MHz and 70 W. On the second step, equal volumes were mixed and sonicated for 30 minutes more; fnally 1 mL of the fnal solution was placed inside stainless steel cylinders of 14 mm diameter mounted on glass plates. Once the solvent was evaporated slowly (casting technique), the flms were placed in an oven at 70[∘] C; once the samples were dried, two parallel lines of silver contacts on the surface were placed using conductive silver paint. In order to compare the efect of diacetylenes in the polymeric structure, ELOpolymer composites were made under the same conditions,

Figure 2: Simplifed scheme for the synthesis of the polymeric matrix.

but because ELO is a liquid monomer, flms of the mixture on glass substrates were cured at 220[∘] C for 2 h.

For electrical characterization, I-V relationship was measured by the two-point technique with an electrometer Keithley 6717A. Samples with 1, 3, 4, 5, 8, 10, and 12% (W/W) of MWCNTs were measured; for this polymer's MWC-NTs composites, an I-V lineal correlation was observed at potentials between 10 and 100 mV. The electrical resistance and the resistivity of the flms were calculated and plotted against MWCNTs percentage (W/W). The thicknesses of the layers were measured using a proflometer Sloan Dektak IIA. Four samples were averaged for each point of the electrical resistance. The percolation theory was applied in order to calculate the percolation threshold, at the onset of the network; electrical resistivity obeys the power law relation:

$$
\sigma = \sigma_m \left(\phi - \phi_c \right)^{\beta}, \tag{1}
$$

where σ is the electrical resistivity of the polymeric compound, ϕ is the filler concentration, ϕ_c is the critical filler concentration or percolation threshold, and β and σ_m are fitting constant. The adjustment was made using Origin 6.0 software. Three free parameters σ_m , ϕ_c , and β were considered, where σ_m is the proportionality constant. Best fitting curves were obtained for $β$ very close to -1.65; then β was fixed to this value and numerical interactions were run again until reaching 0.99 of data correlation.

3. Results and Discussion

ELO was synthesized by a chemoenzymatic reaction (Figure 2). The epoxy rings formation was confirmed by FTIR-ATR, ¹H NMR, and DCS. By FTIR-ATR, the band at 821 cm⁻¹ corresponding to the ν (**C–O**) from epoxy groups was observed (Figure 3(b)), whereas those corresponding to the double bonds at 1650 and 3008 do not (Figure 3(a)). The ¹H NMR spectrum also showed the signals from epoxy

Figure 3: FTIR-ATR spectra of (a) LO, (b) ELO, (c) APELO, and (d) DAELO.

rings between 2.90 and 3.21 ppm, integrating 11.5 hydrogens (Figure 4(b)); these signals allowed calculating the epoxy groups percentage; it was 96% with respect to the signals from the double bonds between 5.23 and 5.42 ppm in the $^1\mathrm{H}$ NMR spectrum of LO integrating 12 hydrogens (Figure 4(a)). The signal of hydrogens from double bonds in the ELO spectrum integrates only 0.56. Molecular weight of LO and ELO was calculated by integration of their respective spectra; for the frst one it was 879 g/mol and for the second one it was 970 g/mol. In DSC an exothermic peak between 205 and 275[∘] C was observed which corresponds to the ring opening of epoxy groups (Figure 5(a)) [29].

FIGURE 4: ¹H NMR 300 MHz, CDCl₃ of (a) LO, (b) ELO, and (c) APELO.

FIGURE 5: DSC and TGA runs of (a) ELO and (b) APELO (solid lines) and DAELO (dashed lines). Heating rate of 20°C/min under nitrogen atmosphere.

ELO was functionalized with PA by an epoxy ring opening reaction catalyzed by ZnCl₂; PA was used as reactant and solvent in order to obtain better results. The higher percent of functionalization (27%) was obtained when the reaction was carried out at 80[∘] C, 20 h, and 20 mol percent of catalyst. The products were characterized by FTIR-ATR and ¹H NMR. In the FTIR-ATR spectra the bands of ν (C≡**C**–**H**) at 3271 cm⁻¹ and ν (**C**≡**C**) at 2121 cm⁻¹ were observed as well as the characteristic band centered around 3450 cm[−]¹ from hydroxyl groups (Figure $3(c)$); the band from epoxy groups was not observed. On the other hand, in the ¹H NMR spectrum of APELO, the signal of methylene from PA appeared as a doublet at 4.68 ppm; $J = 3$ Hz (Figure 4(c)); the integration allowed calculating the percent of functionalization with respect to the epoxy groups; for the highest functionalization (28%) the integral had a value of 3.20, corresponding to 1.6 acetylene groups. When the reaction was carried out for only 5 h, the functionalization was of 26%; therefore, these conditions were used for the APELO synthesis. Results for other conditions are shown in Table 1. Low values should be due to hydroxyl group formation which attack epoxy rings to form ethers. This is evidenced by IR (Figure 3), where the band of ν (**C**–**O**) at 1037 cm⁻¹ is more intense than in ELO and LO spectra.

Cross-linking of acetylenes of PAELO was carried out by the Hay reaction. The product obtained (DAELO) was

Figure 6: SEM micrographs of polymeric composites with (a) 5% and (b) 10% of MWCNTs, 30 kV, and 5000x.

Table 1: Results of the functionalization of ELO with PA at diferent conditions.

Catalyst (%)	Reaction time	Temperature	Functionalization (%)
10	5	80	8
20	3	80	19
20	5	80	26
20	20	80	27
20	5	70	22
20	8	70	20
30	5	70	20

a yellow solid which was characterized by FTIR-ATR, DSC, and TGA. In FTIR, the band of ν (C≡**C–H**) at 3270 cm⁻¹ diminished with respect to PAELO which confrm the C– C coupling (Figure $3(d)$). This coupling improves thermal stability as shown by TGA (Figure 5(b)). The T_{10} for DAELO was 216[∘] C, whereas for PAELO it was 154[∘] C. In DSC an exothermic peak starts at 176 and 187[∘] C for PAELO and DAELO, respectively, owing to their corresponding decomposition.

Once the resin is obtained, the polymeric compounds were prepared according to the experimental section. Chloroform was chosen as solvent due to the fact that it has been used at high MWCNTs concentrations [30]. Interestingly, only one hour of ultrasound was enough to disperse the MWCNTs inasmuch as longer times might cause fractures on CNTs [24]. Dispersion was confrmed by images from SEM (Figure 6) for samples with 5 and 10% at 5000x.

The electrical resistance of DAELO-MWCNTs compounds was measured for samples with 1, 3, 4, 5, 8,10, and 12% (W/W) of MWCNTs at potentials between 10 and 100 mV; in all cases a lineal relation was obtained, indicating an adequate electric contact between the network of MWCNTs and a very good distribution of them in the matrix. It is also noted that a good tendency of resistivity is obtained in the limits of the percolation theory; this is difficult to reach for other particles if the synthesis conditions are not optimized. Resistivity values are shown in Figure 7; these

Figure 7: Resistivity as a function of MWCNTs percentage (W/W), mean value \pm SD ($n = 4$), and their adjusted curve.

were adjusted by percolation threshold equation fnding the critical concentration at 0.99% (W/W) which is similar to other polymer MWCNTs compounds; for example, the percolation threshold for polystyrene-single walled carbon nanotubes composites was at 1.5% (W/W) [31], whereas for polyvinylidene fuoride MWCNTs it was at 0.95% [32]. Although the density of DAELO was slightly lower than ELO, the volumetric fraction of MWCNTs was similar in both compounds. In comparison, the measured compositions of ELO-composites always had a higher value of resistivity and a higher error, indicating a major problem of dispersion of CNTs in that molecule; even if it is true, the resistivity of ELO without CNTs is higher than DAELO; results are given in Table 2. This result could be explained in terms of π interactions between the electrical density of diacetylenes groups (1 per molecule in average) and electrical density of MWCNTs allowing a better dispersion [33]. By tailoring the polymer matrix, it is possible to improve the interactions between the electronic density of the functional groups in the polymers and the CNTs reaching a low critical concentration in the electrical conductive composites.

MWCNTs	ρ (k Ω ·m)	ρ (k Ω ·m)
(W/W)	DAELO-MWCNTs	ELO-MWCNTs
Ω	7.1×10^{7}	1.7×10^{9}
$\mathbf{1}$	$9.08 \times 10^6 + 4.9 \times 10^6$	
3	$1559 + 1143$	
$\overline{4}$	$686 + 196$	
-5	408 ± 19	4697 ± 1642
8	$170 + 31$	$5711 + 3658$
10	$99.7 + 27.5$	3160 ± 944
12	81.6 ± 5.7	3101 ± 1392

Table 2: Comparison of resistivity of composites at diferent percentage of MWCNTs.

Densities of ELO, DAELO, and MWCNTs were 0.99, 0.89, and 1.9 $\rm g/cm^3$, respectively.

4. Conclusions

In this work, the functionalization of ELO with PA was only 27%; however it was possible to make a resin by oxidative coupling of the terminal alkynes; this resin was used as matrix for the polymer MWCNTs composites formation where the introduction of a few diacetylenes groups allows the dispersion of MWCNTs with a short time of sonication. The percolation threshold was at low concentration of conductive particles; these decreased the electrical resistivity in fve orders of magnitude from 1% to 10% in weight of MWCNTs. Because the combination of CNTs and polymers with π electrons had been of interest for electronic devices, we think that these composites may have potential application, for example, in vapor sensors.

Conflict of Interests

The authors declare that there is no conflict of interests regarding the publication of this paper.

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